

# AN ALTERNATIVE SCHEME FOR STIFFENING SRF CAVITIES BY PLASMA SPRAYING

S.Bousson, M.Fouaidy, H.Gassot, T.Junquera, J.Lesrel, IPN Orsay, France

J.L Borne, J.Marini, LAL Orsay

C.Antoine, J.P.Charrier, H.Safa, DSM/DAPNIA/SEA CEA Saclay

## Abstract

Stiffening of bulk niobium SRF cavities is mandatory for reducing the frequency shift induced by Lorentz forces at high accelerating gradients. Experimental and computational data previously reported show that with the actual scheme (i.e. EB welded stiffening rings) the frequency shift of TESLA 9 cells SRF cavities is higher than the cavity bandwidth above  $E_{acc}=28$  MV/m. We propose a new stiffening method, using a Plasma Sprayed Copper Layer (PSCL) onto bulk niobium cavities. As compared to the actual technique, this method offers several advantages (simplicity, reliability...). The first experimental data obtained with monocell cavities produced by this method demonstrate the efficiency of cavities stiffening with plasma spraying. Thermal and mechanical properties measured on niobium samples with a PSCL are also presented. These data will allow us choose the plasma spraying process suitable for achieving the best cavities performances.

## 1 INTRODUCTION

Recent results obtained with 9-cells TESLA cavities point out a new problem for cavity stiffening. The actual EB welded stiffening rings are no more efficient for accelerating field above 28 MV/m, Lorentz forces detuning becoming too important as compared to the cavity bandwidth. As cavities recently reached 33 MV/m [1], stiffening is already a problem and a solution has to be found. A new stiffening method is proposed, based on the coating of bulk Nb cavities by a plasma sprayed copper layer. The coating must be efficient for accelerating fields up to 40 MV/m, which is the ultimate TESLA goal. Thanks to its good thermal conductivity, copper was the best material candidate to avoid cavity performances degradation. Mechanical characteristics of the copper coating could be close to bulk material with a suitable spraying process. As the Young modulus decreases strongly with the porosity, we have to find a spraying process which allows the lowest possible porosity (a few percent). Bond strength and achievable thickness are also very important issues. These properties are essential for choosing the more suitable spraying process.

## 2 THERMAL SPRAYING TECHNIQUES

The different thermal spraying methods can be divided into 3 different kinds [2].

### a) Plasma spraying.

The principle is to create a plasma by an electric arc discharge initiated in a gas (usually Ar/H<sub>2</sub>). The copper powder is injected in the high temperature plasma and the molten particles are sprayed out of the plasma gun. Depending on the spraying environment, different techniques were developed: under air (Atmospheric Plasma Spaying, APS), under inert gas (Controlled Atmosphere Plasma Spraying, CAPS), and under vacuum (Vacuum Plasma Spraying, VPS).

### b) Combustion flame spraying.

The flame spraying (FS) principle is to use the chemical energy of combustion of fuel gas in oxygen to heat up the powder. If the oxygen is at a high pressure, the method is called High Velocity Oxy-Fuel Spraying (HVOF), and when an explosive mixture of oxygen and acetylene is used to post accelerate with the detonation (1-15 detonations per second), it is called Detonation-Gun Spraying (DGS).

### c) Arc Spraying (AS).

Consumable electrodes made by two wires of the coating material are molten by arc heating, and the produced droplet is propelled by compressed gas. Some coating mechanical properties are summarised in Table 1.

Table 1: Main coating properties.

Method	porosity	bond strength	Comments
APS	medium	high	
VPS	low	high	no oxidation
CAPS	medium	high	no oxidation
FS	high	low	
DGS	low	high	pulsed
HVOF	low	high	
AS	high	low	

The first cavities were copper coated using the industrial APS method, and we are now working in a close collaboration with Ecole Nationale Supérieure des Mines de Paris to improve this technique (lower porosity, higher bond strength without bonding layer). Moreover the HVOF and VPS methods are investigated with help of Institut Polytechnique de Sevenans (LERMPS, France).

### 3 RF TESTS ON Nb/Cu CAVITIES

In order to study the feasibility of the new fabrication method, we performed RF tests on a 1.3 GHz monocell cavity before and after the copper coating. Initially, the cavity was 2.5 mm thick, made from RRR 200 Nb sheets, and then stiffened with a 2.5 mm thick copper layer. The coating was made by a "rough APS", a not optimised process, with the use of an intermediate  $\approx 0.2$  mm thick bonding layer (bronze/aluminium alloy) between niobium and copper. The two resulting  $Q_0$  vs  $E_{acc}$  curves (Fig.1) show only a slight reduction of the maximum attainable field (quench), while the  $Q_0$  level is almost not decreased.

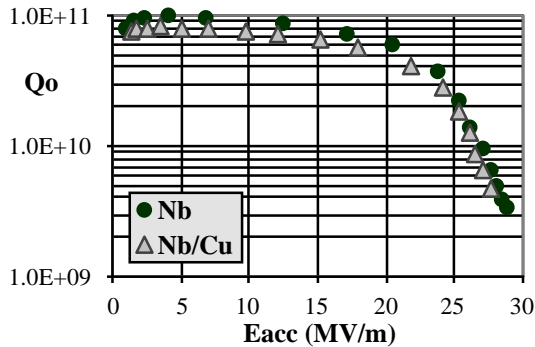


Fig. 1:  $Q_0$  vs  $E_{acc}$  curves before and after copper coating on a 1.3 GHz monocell cavity (C1 02) @ 1.7 K.

During these experiments, the cavity frequency shift induced by Lorentz forces was measured. As theoretically expected, the frequency shift due to Lorentz forces depends quadratically on the accelerating field:  $\Delta f = K \cdot E_{acc}^2$ , where K is a constant. On the Fig. 2, the 35 % decrease of the slope of  $\Delta f$  vs  $E_{acc}^2$  curve (Nb vs Nb with APS Cu coating) gives the stiffening efficiency of the APS copper coating.

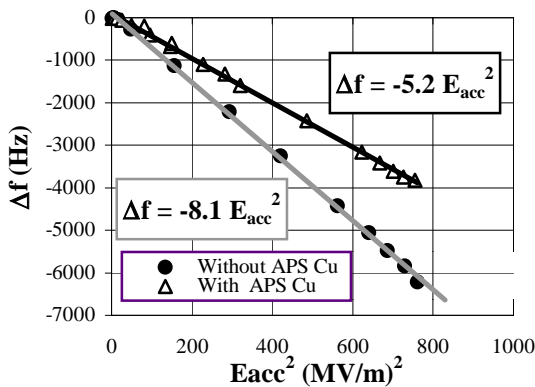


Fig 2 :  $\Delta f$  vs  $E_{acc}^2$  with and without APS Cu.

To carry on the study of the new stiffening method, five 3 GHz cavities were fabricated from RRR 40 Nb sheets of 0.5 mm thickness [3]. The copper deposition process was first tested on the cavity #3, and then two more cavities

(#4 and #5) were RF tested before and after Cu coating by the "rough APS" method. The results summarised in Table 2 show that the cavity performances are almost not modified by the APS Cu coating. Cavities #1 and #2 were tested before stiffening and reached high accelerating fields despite the poor Nb quality ( $E_{acc} = 24.5$  MV/m for cavity #2). A study of several PS methods are currently in progress on samples in order to define the best stiffening method that will be tested on cavity #1 and #2.

Table 2: 3 GHz cavity tests @ 1.8 K.

Cavity number	$E_{acc}$ max before Cu deposition	$E_{acc}$ max with Cu
# 1	12.5 MV/m	to be tested
# 2	24.5 MV/m	to be tested
# 3	not tested	10 MV/m
# 4	16.5 MV/m	16.5 MV/m
# 5	14.5 MV/m	13.5 MV/m

### 4 THERMAL INVESTIGATIONS

The stiffening coating adds a supplementary thermal resistance  $\Delta R_g$  on the overall thermal resistance ( $R_g$ )  $R_g = R_c + R_k + \Delta R_g$ , with  $R_c$  the conductivity term for the niobium and  $R_k$  the Kapitza resistance term (Kapitza resistance is nearly the same for Nb and Cu). Thermal simulations were performed to determine the  $\Delta R_g$  threshold above which the cavity thermal behaviour could be modified (either by a maximum accelerating field decrease or by an effect on the  $Q_0$  level). Both defect free case and defect case were studied. On the figure 4 is plotted the defect free case theoretical  $Q_0$  vs  $E_{acc}$  for bulk Nb ( $\Delta R_g = 0$ ). We have then calculated the  $Q_0$  vs  $E_{acc}$  curves for an increase  $\Delta R_g$  of the overall thermal resistance, which is arbitrarily taken as equal to  $R_k$  and 3  $R_k$ . These two runs simulates a possible increase of  $R_g$  due to the copper coating.

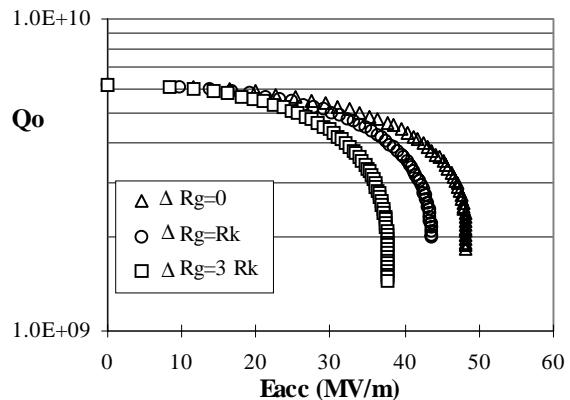


Fig. 4 : Simulation of  $Q_0$  vs  $E_{acc}$  (defect free case) at  $T = 2$  K @ 1.3 GHz with RRR=200 and 2.5mm thick.

The results clearly show that if  $\Delta R_g = R_k$  ( $= 1.4 \cdot 10^{-4}$  W.m<sup>2</sup>/K @ 2 K), the TESLA goal ( $E_{acc} = 40$  MV/m)

could be achieved, whereas if  $\Delta R_g = 3 R_k$ , this goal is not reached. The defect case study leads to the same conclusion, so a threshold on the coating thermal resistance was determined for achieving the stiffening without drawbacks at 40 MV/m.

In order to measure  $\Delta R_g$  due to the copper layer on our cavities, Nb samples were coated by the same "rough APS" process.  $\Delta R_g$  is obtained from the overall thermal resistance difference between a Nb sample and a Cu coated Nb sample [3]. At 1.8 K,  $\Delta R_g$  was found to be  $6.4 \cdot 10^{-4} \text{ Km}^2/\text{W}$  (about 4 times  $R_k$ ). In this result, the effect of the bonding layer (0.2 mm thick bronze/aluminium alloy) was suspected to dominate the thermal resistance, so another Nb sample coated with only the alloy coating was also measured. The result was the same ( $\Delta R_g = 6.2 \cdot 10^{-4} \text{ Km}^2/\text{W}$ ) clearly indicating the very low thermal conductivity of the bonding layer.

### 5 MECHANICAL SIMULATIONS

Numerical simulations have been performed to study the effect of different stiffening schemes using an additional copper layer on a TESLA 9-cells cavity detuning. The basics of the model are to consider a bimetal Nb/Cu cavity (see Fig. 5), each material considered as bulk material perfectly bonded to each other [3].



Fig.5 : Stiffened bimetal Nb/Cu cavity.

This ideal case was used to model different stiffening options, with homogeneous or non-homogeneous (thicker coating at the iris) copper layer, and Lorentz forces detuning @  $E_{acc} = 40 \text{ MV/m}$  was computed for TESLA 9-cells cavities. The results are reported in Table 3. They show that with the actual stiffening scheme (2.5 mm Nb thickness and EB welded stiffening rings), the frequency shift is twice the cavity bandwidth ( $434 \text{ Hz}$  for  $Q_{ext} = 3 \cdot 10^6$ ). Thanks to a non-homogeneous copper coating (2 mm thick layer and 20 mm at the iris), a frequency shift @ 40 MV/m within the cavity bandwidth was obtained.

Table 3: Computed frequency shift for TESLA cavities at  $E_{acc} = 40 \text{ MV/m}$  for different stiffening schemes.

Configuration	$\Delta f$
niobium 2.5 mm unstiffened	-2135Hz
Nb 2.5 mm + EB welded stiffening rings	<b>-863 Hz</b>
Nb 2.5 mm + Cu coating 2 mm	-883 Hz
Nb 2.5 mm + Cu coating 2 mm + iris stiffening (h=20mm)	<b>-358 Hz</b>

The Young modulus of the coating used with the 3 GHz cavity prototypes was estimated to 27 GPa from a simple experiment which consists in measuring the coated cavity deformation versus the applied axial force and comparison with a model calculation. Using this value,

simulations on 9-cells TESLA cavities give a  $\Delta f$  of 616 Hz at 40 MV/m (non-homogeneous scheme), a value higher than the cavity bandwidth, but still an improvement as compared to the 863 Hz obtained with the actual EB welded stiffening rings. Simulations also show that this coating is efficient for accelerating fields up to 33 MV/m. Different theoretical approaches modelled the effect of the porosity on the Young modulus. As the bulk copper Young modulus is 130 GPa, the estimated value (27 GPa) is in the range of expected values of APS coating with porosity lying between 15% and 30%. Mechanical model calculations show that to achieved an effective stiffening at 40 MV/m, it is necessary to have a coating Young modulus of 95 GPa, which corresponds to a porosity of a few percent.

### 6 CONCLUSION AND FUTURE

A new fabrication method for SRF cavities is presented. The principle is to stiffen niobium cavities with a copper layer deposited by thermal spraying. Comparison of RF performances obtained with the first cavities tested before and after copper deposition showed that the maximum accelerating field was not affected, while a stiffening effect was measured. These results demonstrate the interest of the method. But this "rough APS" copper coating is efficient only for  $E_{acc} < 33 \text{ MV/m}$ . Mechanical simulations have proved that porosity have to be less than a few percent. Thermal measurements showed that for this coating at 40 MV/m, the cavity thermal stability should not be affected if the bonding layer is removed. This first study pointed out the important parameters involved in the cavity stiffening. Now, a new program has just started to investigate other spraying process (VPS, HVOF, and a more controlled APS), more suited to our application. After measurements of mechanical and thermal properties on samples, 3 GHz and 1.3 GHz cavities will be fabricated from 1 mm thick Nb sheets of RRR 130 and then stiffened with the copper layer.

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